ORIGINAL PAPER

Electrochemical formation and properties of two-component films of transition metal complexes and C_{60} or C_{70}

Emilia Grodzka · Joanna Grabowska · Monika Wysocka-Żołopa · Krzysztof Winkler

Received: 1 November 2007/Revised: 4 December 2007/Accepted: 4 December 2007/Published online: 8 January 2008 © Springer-Verlag 2007

Abstract The electrochemically active polymers have been formed during electro-reduction carried out in solution containing fullerenes, C₆₀ or C₇₀, and transition metal complexes of Pd(II), Pt(II), Rh(III), and Ir(I). In these films, fullerene moieties are covalently bounded to transition metal atoms (Pd and Pt) or their complexes (Rh and Ir) to form a polymeric network. All films exhibit electrochemical activity at negative potentials due to the fullerene cages reduction process. For all studied metal complexes, yields of formation of films containing C₇₀ are higher than yields of electrodeposition of their C₆₀ analogs. C₇₀/M films also exhibit higher porosity in comparison to C_{60}/M layers. The differences in film morphology and efficiency of polymer formation are responsible for differences in electrochemical responses of these films in acetonitrile containing supporting electrolyte only. C_{70}/M films shows more reversible voltammeric behavior in negative potential range. They also show higher potential range of electrochemical stability. Processes of film formation and electrochemical properties of polymers depend on the transition metal ions or atoms bonding fullerene cages into polymeric network. The highest efficiency of polymerization was observed for fullerene/Pd and fullerene/Rh films. In the case of fullerene/Pd films, the charge transfer processes related to the fullerene moieties reduction in negative potential range exhibit the best reversibility among all of the studied systems. Capacitance performances of C₆₀/Pd and C₇₀/Pd films deposited on the porous Au/quartz electrode were also compared. Capacitance properties of both films are signifi-

E. Grodzka · J. Grabowska · M. Wysocka-Żołopa ·

Institute of Chemistry, University of Białystok, Piłsudskiego 11/4, 15-443 Białystok, Poland e-mail: winkler@uwb.edu.pl cantly affected by the conditions of electropolymerization. Only a fraction of the film having a direct contact with solution contributes to pseudocapacitance. Capacitance properties of these films also depend on the size of cations of supporting electrolyte. The C₇₀/Pd film exhibits much better capacitance performance comparison to C₆₀/Pd polymer.

Keywords Fullerenes · Fullerene polymers · Electropolymerization · Transition metal complexes

Introduction

The new electrochemically active materials have been of considerable interest in recent years due to their potential application as electroactive materials in batteries [1], super-capacitors [2], sensors [3, 4], or electrochromic devices [5]. Recently, considerable attention has been paid to the preparation and properties of polymers containing fullerenes [6–12]. The incorporation of the pseudo-spherical π -electron cages of fullerenes into solid-phase can significantly modify electronic, magnetic, and optical properties of materials.

Fullerenes can form homopolymers through [2+2] cycloaddition [13–18]. These systems are relatively unstable. Fullerenes can be also incorporated into main polymeric chain to form a "pearl necklace" structure [19–22] or to the side chains of polymers as pendant substituents [23–26]. So far, chemical polymerization is mostly used for the preparation of fullerene-containing polymers. However, the rate of chemical polymerization is difficult to control. Moreover, the structure of the resulting systems is poorly defined in many cases. Using electrochemical polymerization instead can remove some of these disadvantages.

Recently, we have focused on the development of novel electroactive polymers based on fullerenes [27-37]. The

K. Winkler (🖂)

electrochemical reduction of fullerene epoxides, C₆₀O and C₇₀O, results in deposition of polymeric films onto electrode surfaces [27-29]. Two-component films of fullerenes or their derivatives and transition metal complexes are formed on the electrode during reduction curried out in solution containing fullerenes and complexes of such transition metals as palladium [30-33], platinum [34], rhodium [30], or iridium [30]. In the resulting films, the fullerene moieties are believed to be bonded to the metal centers in η^2 fashion to form a polymeric network. Polymers of C_{60} and transition metal complexes, C_{60}/M , exhibit electrochemical activity in the negative potential range due to the reduction of the fullerene moieties present. The process of film reduction is accompanied by the transport of cations from the supporting electrolyte into the film [31]. Therefore, these films exhibit n-doped properties. Redox-active films of transition metals and fullerene derivatives containing electron-donating groups, 2'-ferrocenylpyrrolidino-[3',4';1,2][C₆₀]fullerene and C₆₀ with a covalently linked zinc meso-tetraphenylporphyrin, can be also synthesized under electrochemical conditions [35–37]. These films can undergo both *p*- and *n*-doping. At negative potentials, reduction of fullerene moieties occurs. In positive potential range, the processes of ferrocene or zinc meso-tetraphenylporphyrin oxidation take place.

Recently, it was also shown that C_{60} /Pd polymer exhibits very good capacitance performance [38]. Capacitance properties of this polymer depend on the condition of polymer deposition, solvent, and supporting electrolyte.

So far, most of the work was focused on the study of formation and properties of $[C_{60}]$ fullerene-based electroactive materials. In this paper, the processes of electropolymerization of two component films of transition metal complexes and $[C_{60}]$ fullerene, C_{60}/M , and $[C_{70}]$ fullerene, C_{70}/M , are compared. The electrochemical properties of these films are also investigated. A special attention is paid to comparison of capacitance properties of C_{60}/Pd and C_{70}/Pd films.

Experimental

Palladium(II) acetate (98%), Pd(ac)₂, (Aldrich), di- μ -chlorodichlorobis(ethylene)-diplatinum(II) (97%), [Pt(μ -Cl)Cl (C₂H₄)]₂, (ABCR GmbH, Karlsruhe, Germany), rhodium (II) trifluoroacetate dimer, [Rh(CF₃COO)₂]₂, (Aldrich), [C₆₀] fullerene (99.5+%; MER Corporation, Tuscon, Arizona) and [C₇₀]fullerene (98.5%; Southern Chemical Group) were used as received. Iridium complex, IrCl(CO)₂(*p*-toluidine), was synthesized according to procedure described elsewhere [39]. The supporting electrolytes, tetra(ethyl)ammonium perchlorate (Acros Organics), tetra(*n*-butyl)ammonium perchlorate (Fluka) and tetra(*n*-hexyl)ammonium perchlorate (Alfa Aesar), were dried under vacuum for 24 h before use. Acetonitrile (99.9%) and toluene (99.8%) were used as received from Aldrich Chemical.

Voltammetric experiments were performed on a potentiostat/galvanostat Model 283 (EG&G Instruments) with a three-electrode cell. A gold disk with a diameter of 1.5 mm (Bioanalytical Systems) was used as the working electrode. Before the experiment, the electrode was polished with fine carborundum paper and then with a 0.5- μ m alumina slurry. Subsequently, the electrode was sonicated in water to remove traces of alumina from the gold surface, washed with water, and dried. A silver wire immersed in 0.010 M silver perchlorate and 0.09 M tetra(*n*-butyl)ammonium perchlorate in acetonitrile that was separated from the working electrode by a ceramic tip (Bioanalytical Systems) served as the reference electrode. The counter electrode was a platinum tab with an area of about 0.5 cm².

Simultaneous voltammetric and piezoelectric microgravimetry experiments were carried out with a home-built potentiostat and an electrochemical quartz crystal microbal-



Fig. 1 Multicyclic voltammograms for **a** 0.25 mM C₆₀ and 0.85 mM Pd(ac)₂, and **b** 0.25 mM C₇₀ and 0.85 mM Pd(ac)₂ in acetonitrile/ toluene (1:4, *v:v*) containing 0.1 M tetra(*n*-butyl)ammonium perchlorate recorded at gold disk Au (1.5 mm diameter). The sweep rate was 100 mV s⁻¹

Fig. 2 Multicyclic voltammograms and curves of the frequency changes vs potential simultaneously recorded at the same Au/quartz electrode in acetonitrile/toluene (1:4, *v:v*) containing 0.1 M tetra(*n*-butyl) ammonium perchlorate and **a** 0.25 mM C₆₀ and 0.85 mM Pd (ac)₂, and **b** 0.25 mM C₇₀ and 0.85 mM Pd(ac)₂. The sweep rate was 25 mV s⁻¹



ance, EQCM 5510 (Institute of Physical Chemistry, Warsaw, Poland). Plano-convex quartz crystals were used. The 14 mm diameter AT-cut, plano-convex quartz crystals with a 5-MHz resonant frequency were obtained from Omig (Warsaw, Poland). A 100-nm gold film, which was vacuum deposited on the quartz crystal, served as the working electrode. The projected region of this Au electrode was 5 mm in diameter. The area of the circuit center spot and two contacting radial strips was 0.24 cm². Unpolished quartz crystals were used for better adherence of the film. The sensitivity of the mass measurement calculated from the Saurbrey equation was 17.7 ng Hz⁻¹ cm⁻². A mass range within which Saurbrey equation is valid has been determined using the process of polypyrrole electrodeposition. Polypyrrole was formed under potentiostatic conditions. A linear relationship between the charge of pyrrole oxidation and mass of polymer deposited on the electrode surface was observed for mass as large as 100 μ g cm⁻². The Au/quartz electrode with surface area of 0.24 cm^2 was used for study of capacitance properties of polymers.

Secondary electron scanning electron microscopy images were obtained with the use of an S-3000N instrument of Hitachi. The accelerating voltages for the electron beam were 10 and 20 keV, and the average working distance was 5 mm.

The C_{60}/M and C_{70}/M films were prepared by electroreduction of an acetonitrile/toluene (1:4, *v*:*v*) solution that contained the fullerene and transition metal complexes, and the supporting electrolyte, 0.10 M tetra(*n*-butyl)ammonium perchlorate. Films were grown under cyclic voltammetric conditions mostly at a potential sweep rate of 100 mV/s in the potential range from 100 to -1,450 mV. The electrochemical properties of the film were studied in a solution of the chosen solvent containing only the supporting electrolyte. In this case, the electrode covered with the film was removed from the growth solution, rinsed several times with an acetonitrile/toluene (1:4, *v*:*v*) solution and then placed in

solution containing 0.10 M of supporting electrolyte. The modified electrode was allowed to equilibrate for 10 min, while degassing with argon in a fresh solution before electrochemical measurements were performed.



Fig. 3 Dependences of the mass of C_{70} /Pd (Δm) deposited on the Au/ quartz electrode under EQCM conditions on the charge of reduction (*Q*). The grown acetonitrile/toluene (1:4, *v:v*) solution contained 0.25 mM C_{70} , 0.85 mM Pd(ac)₂ and 0.1 M tetra(*n*-butyl)ammonium perchlorate. The sweep rate was 25 mV s⁻¹. Reduction charge was calculated by integration of voltammetric curve shown *inset. Arrows* indicate potentials corresponding to the charges of the change of $Q-\Delta m$ slope



Fig. 4 Dependences of the mass of Au/quartz electrode on the cycle number during C_{60} /Pd (**a**) and C_{70} /Pd (**b**) film deposition in acetonitrile/toluene (1:4, *v*:*v*) containing 0.1 M tetra(*n*-butyl)ammonium perchlorate, **a** 0.25 mM C_{60} and **b** 0.25 mM C_{70} . Concentration of Pd(ac)₂ was 1 mM (*empty circles*), 2 mM (*empty diamonds*), 2.5 mM (*filled squares*), 3 mM (*filled diamonds*), and 4 mM (*filled circles*)

Results and discussion

 C_{60}/Pd and C_{70}/Pd polymers The C₆₀/Pd and C₇₀/Pd polymeric thin films are formed during reduction carried out in acetonitrile/toluene (1:4, *v*:*v*) mixture containing Pd (ac)₂ and fullerene. Voltammograms of C₆₀/Pd and C₇₀/Pd polymer formation are compared in Fig. 1. Upon repeated scanning of the potential between 100 and -1,700 mV with a scan rate 100 mV/s, an increase of the current in the potential range for fullerene moiety reduction is seen for two systems. For the processes of both polymer deposition, qualitative and quantitative differences are observed. Peak currents related to the reduction of fullerene moieties are much better defined for the process of C₇₀/Pd deposition. However, currents observed for the process of C₆₀/Pd formation are higher than currents recorded in the solution containing C₇₀. Figure 2 shows results of electrochemical quartz crystal microbalance (EQCM) studies of the processes of C₆₀/Pd and C70/Pd film electrodeposition on Au/quartz electrode. The frequency decreases in the cathodic cycle in the potential range of palladium cation reduction due to the polymer deposition process. Three ranges with different slopes corresponding to the reduction of Pd(II) and two fullerene reduction steps are observed on $E-\Delta f$ curves. A small decrease of frequency in the positive scan direction is related to the film oxidation followed by egress of the supporting electrolyte cations from the polymer to the solution. Frequency changes observed during C70/Pd deposition are higher then the frequency changes obtained for the process of C₆₀/Pd electropolymerization. These results are rather puzzled in respect to voltammograms of films formation which are shown in Fig. 1 as multicyclic voltammograms of polymer electrodeposition indicate higher efficiency of C₆₀/Pd film formation.

Figure 3 shows dependences of the mass of C_{70} /Pd polymer, Δm , deposition at the electrode surface on the charge of reduction, Q, obtained on the base of results presented in Fig. 2. Reduction charge was obtained by integration of voltammogram. This graph can be divided into



Potential (mV)

Fig. 5 Cyclic voltammograms of C_{60} /Pd (**a**) and C_{70} /Pd film (**b** and **c**) in acetonitrile containing 0.10 M tetra(*n*-butyl)ammonium perchlorate (**a** and **b**) and 0.10 M tetra(ethyl)ammonium perchlorate (**c**) recorded at Au (1.5 mm). The sweep rate was 100 mV s⁻¹. The C_{60} /Pd and C_{70} /Pd films were grown under cyclic voltammetry conditions in acetonitrile/toluene (1:4, *v*:*v*) containing 0.10 M tetra(*n*-butyl)ammonium perchlorate, 0.25 mM C_{60} or C_{70} , and 0.85 mM Pd(ac)₂

three parts. Initially, for the low charges, a linear $Q-\Delta m$ relation with a high slope equal to 2.36 µg/mC is observed. This slope is very close to the slope of $Q - \Delta m$ relation obtained for the process of palladium deposition from acetonitrile/toluene (1:4, v:v) mixture containing palladium acetate and supporting electrolyte only. Potential of Pd(ac)₂ reduction process is about 50 mV less negative than the potential of C₇₀ reduction. In the range of higher reduction charges, corresponding to the fullerene first reduction step, the slope of $Q - \Delta m$ relation becomes low and equal to 0.76 μ g/mC. It is obvious that formation of C₇₀/Pd polymer on the electrode surface inhibits the process of palladium deposition and decrease of slope of $Q-\Delta m$ relation occurs. At higher charges corresponding to the second fullerene reduction step, a very significant increase of slope of $Q-\Delta m$ relation to 1.65 µg/mC is observed. These large changes in slope cannot be explained by higher doping of the polymer with supporting electrolyte cations. They indicate that the

Fig. 6 SEM images of films formed on gold foil during 20 (a and c) or 40 (b and d) voltammetric cycle in the potential range +100 to -1,450 mV in acetonitrile/toluene (1:4, *v*.*v*) containing 0.1 M tetra(*n*-butyl) ammonium perchlorate and a and b 0.25 mM C₆₀ and 2.00 mM Pd(ac)₂, and c and d 0.25 mM C₇₀ and 2.00 mM Pd(ac)₂ rate of polymerization is also influenced by the charge of fullerene cage. Probably, increase of negative charge on C_{60} and C_{70} during their reduction favors the bonding of palladium to fullerene and rate of polymerization increases.

Similar behavior was observed for the process of C_{60} /Pd electropolymerization. The slope of initial part of $Q-\Delta m$ relation is 2.38 µg/mC. Next, the polymer is deposited, and the slope drops to 0.51 µg/mC. A lower value of the slope of this linear part of the graph in comparison to the value obtained for C_{70} /Pd system indicates lower efficiency of C_{60} /Pd film formation. In the potential range of second C_{60} reduction step, the slope of $Q-\Delta m$ relationship increases to 1.91 µg/mC.

In Fig. 4, the changes of mass of the electrode as a function of a scan number are shown for solutions containing different concentration of $Pd(ac)_2$ and the same concentration of C_{60} and C_{70} . The mass of the polymer deposited in single voltammetric cycle was calculated from



Fig. 7 Cyclic voltammograms of a C₆₀/Pd and b C₇₀/Pd films in acetonitrile containing 1 0.10 M tetra(ethyl)ammonium perchlorate, 2 0.10 M tetra(nbutyl)ammonium perchlorate, and 3 0.10 M tetra(n-hexyl) ammonium perchlorate in the potential range of fullerene cage reduction recorded at Au/quartz electrode. The sweep rate was 100 mV s⁻¹. The C₆₀/Pd and C70/Pd films were grown under cyclic voltammetry conditions in acetonitrile/toluene (1:4, v:v)containing 0.10 M tetra(n-butyl) ammonium perchlorate, 0.25 mM C₆₀ or C₇₀ and 2.0 mM Pd(ac)₂



a difference in frequency on the beginning of negatively going scan and at the end of positively going scan. For low concentration of Pd(ac)₂ (lower than about 2 mM for a solution containing C_{60} and 2.5 mM for a solution containing C70), the amount of C60/Pd and C70/Pd film deposited on the electrode surface in each cycle decreases with the increase in the scan number and becomes very small for scan number higher than 10. It indicates that the process of polymer deposition under these conditions becomes inhibited by the polymeric layer deposited on the electrode surface. For low concentration of $Pd(ac)_2$, the mass of polymer deposited in a solution containing C₇₀ is about 1.5 times higher than the mass of C₆₀/Pd polymer deposited from a solution containing fullerene C_{60} . In a solution containing a higher concentration of palladium acetate, an almost linear increase of the mass of the electrode with the scan number is observed even for higher scan numbers. This effect is probably related to simultaneous deposition of fullerene polymer and of metallic particles of palladium [40].

Results presented in Fig. 4 were obtained assuming that according to the Saurbrey equation, only the mass changes of the electrode are responsible for frequency changes of the quartz crystal. In the case of polymeric material, also other factors, like roughness and porosity of polymer and their viscosity and viscoelasticity may contribute to the frequency changes of the crystal [41]. These effects are more significant and can contribute to the mass changes presented in Fig. 4 for porous fullerene/Pd films formed in solutions containing a low concentration of Pd(ac)₂.

The voltammetric behavior of C_{60} /Pd and C_{70} /Pd films in potential range 0 to -1,500 mV in acetonitrile solution containing only supporting electrolyte is compared in Fig. 5. For C_{70} /Pd film, voltammograms were recorded in solution

containing tetra(ethyl)ammonium and tetra(*n*-butyl)ammonium perchlorate. In this potential range, two pairs of voltammetric peaks, R_1/O_1 and R_2/O_2 , corresponding to the two consecutive one-electro reduction steps of C_{60} and C_{70} cages are observed. The voltammetric current recorded for C_{70} /Pd film is much higher than current of C_{60} /Pd film formation. Also, the voltammeric peaks of C_{70} cage reduction are much better pronounced. The C_{70} /Pd film reduction current depends on the size of cation of supporting electrolyte indicating that the process of film reduction is accompanied by the transport of cations from solution into

Table 1 Specific capacitance of the C_{60} /Pd and C_{70} /Pd films in acetonitrile solutions containing different tetra(*n*-alkyl)ammonium perchlorates

Concentration of precursors	Supporting electrolyte (0.1 M)	Specific capacitance (F g ⁻¹)
1.0 mM Pd(ac) ₂ + 0.25 mM C ₆₀ 2.0 mM Pd(ac) ₂ + 0.25 mM C ₆₀ 2.0 mM Pd(ac) ₂ + 0.25 mM C ₆₀ 2.0 mM Pd(ac) ₂ + 0.25 mM C ₆₀ 3.0 mM Pd(ac) ₂ + 0.25 mM C ₆₀ 1.0 mM Pd(ac) ₂ + 0.25 mM C ₇₀ 2.0 mM Pd(ac) ₂ + 0.25 mM C ₇₀ 2.0 mM Pd(ac) ₂ + 0.25 mM C ₇₀ 3.0 mM Pd(ac) ₂ + 0.25 mM C ₇₀ 3.0 mM Pd(ac) ₂ + 0.25 mM C ₇₀	$(n-Bu)_4NClO_4$ $(Et)_4NClO_4$ $(n-Bu)_4NClO_4$ $(n-Hx)_4NClO_4$ $(n-Bu)_4NClO_4$ $(n-Bu)_4NClO_4$ $(Et)_4NClO_4$ $(n-Bu)_4NClO_4$ $(n-Bu)_4NClO_4$ $(n-Bu)_4NClO_4$	$\begin{array}{c} 240^{a} \\ 145^{a} \\ 105^{a} & 130^{b} & 195^{c} \\ 70^{a} \\ 85^{a} \\ 270^{a} \\ 185^{a} \\ 125^{a} & 160^{b} \\ 80^{a} \\ 125^{a} \end{array}$

Conditions of polymer deposition:

 $^{^{\}rm a}$ Twenty cycles in the potential range from 100 to –1,450 mV; sweep rate was 100 $\rm mVs^{-1}$

 $^{^{\}rm b}$ Ten cycles in the potential range from 100 to –1,450 mV; sweep rate was 100 mVs $^{-1}$

 $^{^{\}rm c}$ Five cycles in the potential range from 100 to –1,450 mV; sweep rate was 100 mVs $^{-1}$

Fig. 8 Cyclic voltammograms of a C₆₀/Pd and c C₇₀/Pd film in acetonitrile containing 0.10 M tetra(n-butyl)ammonium perchlorate recorded at Au/quartz electrode. The sweep rate was 1 20, 2 50, 3 100, and 4 200 mV s^{-1} . Dependence of the pseudocapacitive current on the sweep rate for (b) C_{60}/Pd and (d) $C_{70}/$ Pd. Films were grown under cyclic voltammetry conditions in acetonitrile/toluene (1:4, v:v)containing 0.10 M tetra(n-butyl) ammonium perchlorate, 0.25 mM C_{60} or C_{70} , and 2.0 mM Pd(ac)_2



the film. The charge related to the film reduction process is much higher in acetonitrile containing tetra(ethyl)ammonium perchlorate than the charge of C_{70} /Pd electroreduction in the presence of bigger tetra(*n*-butyl)ammonium cations. A similar behavior was observed for C_{60} /Pd film [31].

Current recorded in the potential range of fullerene cage reduction is much higher in the case of C_{70} /Pd film than this one observed for C_{60} /Pd. This effect can be related to the higher efficiency of C_{70} /Pd film formation (Fig. 2). Structural differences of both films may be another reason of such behavior. The morphology of C_{70} /Pd and C_{60} /Pd films are compared in Fig. 6. The C_{70} /Pd film shows higher porosity. Such structure may be easier penetrated by solvent and supporting electrolyte ions during the film reduction leading to higher current.

It was shown that C₆₀/Pd film is a very good electrochemical supercapacitor [38]. The comparison of voltammetric responses of C₆₀/Pd and C₇₀/Pd film (Fig. 5) indicate that capacitance properties of C70/Pd should even be a better electrochemical capacitor. A comparison of capacitance properties of C70/Pd and C60/Pd films were done for films deposited on the rough surface of gold deposited on unpolished quartz crystal. In this case, the polymer sticks better to the surface of the electrode allowing for an easy transfer of the electrode covered with polymer from one solution to the other. It was also observed that increase of the potential range of formation toward the potential of fullerene second reduction step leads to better mechanical and electrochemical stability of the film, particularly for polymers formed in solution containing low concentration of Pd (ac)₂. Therefore, in this paper, capacitance properties were investigated for films deposited under voltammetric conditions in potential range $100 \div -1,450$ mV.

Voltammograms obtained for C₇₀/Pd and C₆₀/Pd polymeric films in acetonitrile containing different tetra(*n*-alkyl) ammonium perchlorates are shown in Fig. 7. Voltammograms recorded in the potential range of films reduction show pseudorectangular cathodic and anodic profile that are the mirror image of one another. Such behavior is typical for ideal capacitor. The departure from the ideal rectangular shape is related to the faradaic process of film reduction and reoxidation. The voltammeric behavior in the potential range



Fig. 9 Multicyclic voltammograms for **a** 0.27 mM C₆₀ and 1.10 mM [Pt(μ -Cl)Cl(C₂H₄)]₂ and **b** 0.27 mM C₇₀ and 1.10 mM [Pt(μ -Cl)Cl (C₂H₄)]₂ in acetonitrile/toluene (1:4, *v*:*v*) containing 0.1 M tetra(*n*-butyl)ammonium perchlorate recorded at Au (1.5 mm). The sweep rate was 50 mV s⁻¹



Fig. 10 Cyclic voltammograms of a C₆₀/Pt and b C₇₀/Pt film in acetonitrile containing 0.10 M tetra(ethyl)ammonium perchlorate recorded at Au (1.5 mm). The sweep rate was 50 mV s⁻¹. The C₆₀/Pt and C₇₀/Pt films were grown under cyclic voltammetry conditions in acetonitrile/toluene (1:4, *v:v*) containing 0.10 M tetra(*n*-butyl) ammonium perchlorate, 0.27 mM C₆₀ or C₇₀, and 1.05 mM [Pt(μ -Cl) Cl(C₂H₄)]₂

of pseudocapacitance is less reversible for C_{70} /Pd film. Specific capacitance, C_s , can be calculated by the integration of cyclic voltammograms according to the equation:

$$C_{\rm s} = \frac{\int i \, dt}{\Delta E \, m} \tag{1}$$

where *i* is the pseudocapacitance current, ΔE is the potential range of integration, and *m* mass of the polymer deposited on the electrode surface. Values of specific capacitances obtained this way are collected in Table 1. Relatively large values of specific capacitance were obtained for both films. It locates them among good supercapacitors. However, specific capacitances of the C₆₀/Pd films are significantly lower than reported in earlier paper [38]. This effect is related to the different conditions of film formation. In this paper, polymers were deposited in relatively high potential range. Therefore, the amount of polymer deposited on the electrode surface in each cycle is relatively large. The mass of C₆₀/Pd film deposited in 20 voltammetric cycles from solution containing 3.5 mM Pd(ac)₂ and 0.25 mM C₆₀ is about 75 μ g cm⁻¹. For comparison, the limiting mass of C₆₀/Pd layers studied earlier [38] was about 25 μ g cm⁻¹. It was shown, that at negative potentials, only a fraction of the polymer is reduced [33]. Therefore, for thicker films, the fraction of electrochemically inactive film increases, and decrease in the specific capacitance is observed. This conclusion is supported by results presented in Table 1. A decrease of number of voltammeric cycles of polymer formation results in increase of film specific capacitance. Also, a very strong effect of grown solution composition on the film specific capacitance is observed. Both effects are related to the changes of the amount of polymer deposited on the electrode surface. For thinner C_{60} /Pd and C_{70} /Pd films obtained during deposition from solution with low concentration of Pd(ac)₂ or for low number of voltammetric scans, values of specific capacitance are close to earlier published data.

Both, faradaic and capacitance components of the current depend linearly on the sweep rate. Therefore, a linear dependence of pseudocapacitance current on the sweep rate should be expected. Exemplary voltammograms recorded for different sweep rates and relation of current as a function of sweep rate are shown in Fig. 8 for C_{60} /Pd and C_{70} /Pd film. The specific capacitance, C_s , calculated from the slope of *i*–*v* linear relation was found to be equal to 155 and 85 F g⁻¹ for



Fig. 11 Multicyclic voltammograms for **a** 0.25 mM C₆₀ and 0.85 mM [Rh(CF₃COO)₂]₂, and **b** 0.25 mM C₇₀ and 0.85 mM [Rh(CF₃COO)₂]₂ in acetonitrile/toluene (1:4, v:v) containing 0.1 M tetra(*n*-butyl) ammonium perchlorate recorded at Au (1.5 mm). The sweep rate was 100 mV s⁻¹

 C_{70} /Pd and C_{60} /Pd film, respectively. In the case of C_{70} /Pd film, a departure from the reversible capacitance behavior is observed for higher sweep rates.

 C_{60}/Pt and C_{70}/Pt polymers Multicyclic voltammograms recorded in acetonitrile/toluene (1:4, *v:v*) solution containing [Pt(μ -Cl)Cl(C₂H₄)]₂ and C₆₀ or C₇₀ are shown in Fig. 9. Also in this case, the increase of the current in the following cycles, due to the redox-active film formation is observed. Similar to the C₆₀/Pd and C₇₀/Pd films, fullerene cages are bounded through platinum atoms in η^2 fashion [34]. However, the addition effect, a shift of reduction potential toward values that are more negative with the increase of scan number is observed. This effect is stronger for C₆₀/Pt film. Such behavior indicates that the polymeric phase deposited on the electrode surface inhibits the process of platinum complex and fullerene reduction. Figure 10 shows voltammograms recorded for electrode covered with C_{60} /Pt and C_{70} /Pt in solution acetonitrile containing only a supporting electrolyte. Similar to palladium-containing polymers, both films are electrochemically active at negative potentials due to the fullerene cages reduction. The shape of voltammograms and slope of current–potential relation at potentials of the film reduction indicate that electrode processes of C_{60} /Pt and C_{70} /Pt films are also much less stable in comparison to their palladium polymers. At potentials more negative than about -1,500 mV, a slow decomposition of films is observed upon cyclic potential scanning.

 C_{60}/Rh and C_{70}/Rh polymers Multicyclic voltammograms of both fullerene and rhodium film formation are shown in Fig. 11. These voltammograms and results of EQCM study

Fig. 12 SEM images of films formed on gold foil during 20 (**a** and **c**) or 40 (**b** and **d**) voltammetric cycle in the potential range +100 to -1,450 mV in acetonitrile/toluene (1:4, *v:v*) containing 0.1 M tetra(*n*-butyl) ammonium perchlorate and **a** and **b** 0.25 mM C₆₀ and 0.85 mM [Rh(CF₃COO)₂]₂, and **c** and **d** 0.25 mM C₇₀ and 0.85 mM [Rh(CF₃COO)₂]₂





Fig. 13 Cyclic voltammograms of C₆₀/Rh film in acetonitrile containing **a** 0.10 M tetra(ethyl)ammonium perchlorate, **b** 0.10 M tetra(*n*-butyl)ammonium perchlorate, and **c** 0.10 M tetra(*n*-hexyl) ammonium perchlorate recorded at Au (1.5 mm). The sweep rate was 100 mV s⁻¹. The C₆₀/Rh film was grown under cyclic voltammetry conditions in acetonitrile/toluene (1:4, *v*:*v*) containing 0.10 M tetra(*n*-butyl)ammonium perchlorate, 0.25 mM C₆₀ and 0.85 mM [Rh (CF₃COO)₂]₂

show that the efficiency of C_{60} /Rh film formation is higher. Both films exhibit similar surface morphology (Fig. 12). Thin films are relatively flat with numerous small outcroppings on the surface. Thicker films are much more porous. However, the porosity of C_{70} /Rh is bigger than the porosity observed for C_{60} /Rh film. The structure of the polymer is a very important factor that determines the redox properties of the film. In the case of a sufficiency porous layer, the solid phase is easily penetrated by the solvent. Also, a transport of ions from the solution into the polymer takes place more easily.

The porosity factor is probably responsible for the differences in voltammeric responses of C_{60} /Rh and C_{70} /Rh films in acetonitrile containing different supporting electrolytes. Voltammograms presented in Figs. 13 and 14 show electrochemical properties of C_{60} /Rh and C_{70} /Rh, respectively. Voltammetric response of both films depends on the size of supporting electrolyte cations indicating that the process of film reduction is accompanied by the transport of cations from solution into the film. Supporting electrolyte cations affect the charge of film reduction, reversibility of the polymer redox processes, and film



Fig. 14 Cyclic voltammograms of C_{70}/Rh film in acetonitrile containing **a** 0.10 M tetra(ethyl)ammonium perchlorate, **b** 0.10 M tetra(*n*-butyl)ammonium perchlorate, and **c** 0.10 M tetra(*n*-hexyl) ammonium perchlorate recorded at Au (1.5 mm). The sweep rate was 100 mV s⁻¹. The C_{70}/Rh film was grown under cyclic voltammetry conditions in acetonitrile/toluene (1:4, *v*:*v*) containing 0.10 M tetra(*n*-butyl)ammonium perchlorate, 0.25 mM C_{70} , and 0.85 mM [Rh (CF₃COO)₂]₂

stability. Both films exhibit higher stability in solution containing small tetra(ethyl)ammonium cation. Under cyclic voltammetric conditions, C_{70} /Rh film is stable up to -2.3 V. The potential range of this polymer stable voltammetric behavior is relatively broad comparison to the stability of C_{60} /Pd film [42]. The decomposition of C_{60} /Rh film starts at potentials more negative than about -2.0 V. In a solution containing bigger tetra(*n*-butyl)ammonium and tetra(*n*-hexyl) ammonium cations, both polymers are less stable and decompose at potentials more negative than about -1.8 V. It was postulated that unstable behavior of fullerene polymers at negative potential is related to the structural changes of the film during cation incorporation and repulsion between negatively charged fullerene cages [31, 32].

The C_{70} /Rh film exhibits also much better reversibility of electrochemical reduction processes, particularly in solution containing tetra(ethyl)ammonium cations (Fig. 14a). In the studied potential range, the reduction peaks related to the first three reduction steps of fullerene cage become better pronounced in comparison to the reduction peaks recorded for C_{60} /Rh film (Fig. 13a).



Fig. 15 Multicyclic voltammograms for **a** 0.25 mM C₆₀ and 0.85 mM IrCl(CO)₂(*p*-toluidine), and **b** 0.25 mM C₇₀ and 0.85 mM IrCl (CO)₂(*p*-toluidine) in acetonitrile/toluene (1:4, *v:v*) containing 0.1 M tetra(*n*-butyl)ammonium perchlorate recorded at Au (1.5 mm). The sweep rate was 100 mV s⁻¹

C₆₀/Ir and C₇₀/Ir polymers Voltammograms of C₆₀/Ir and C_{70} /Ir film formation are shown in Fig. 15. In the potential range of polymer formation, iridium complex, Ir(CO)₂Cl(ptoluidine), is redox inactive [30]. Currents recorded at negative potentials are related to the reduction of fullerenes from solution and film reduction processes. Voltammograms of film deposition show significant differences. In the studied potential range, two reduction steps of the films formed on the electrode surface are observed. Currents corresponding to the first reduction step of fullerene involved in polymeric network, R_1 , are similar for both polymers. The broad reduction peak R_2 , observed on the voltammograms of C_{60} /Ir film formation, is responsible for the second reduction step of C_{60} . In the case of C_{70} /Ir film, this peak is much sharper and better pronounced. The second C70/Ir reduction step occurs at less negative potentials than the second step of C₆₀/Ir reduction. In anodic cycle, currents related to two oxidation steps, O₁ and O₂, are observed.

A similar behavior is observed for C_{60} /Ir and C_{70} /Ir films in acetonitrile solution containing only supporting electrolyte (Fig. 16). Also in this case, peaks related to the fullerene cage reduction steps are better separated and pronounced for C_{70} /Ir film.



Fig. 16 Cyclic voltammograms of a C₆₀/Ir and b C₇₀/Ir films in acetonitrile containing 0.10 M tetra(*n*-butyl)ammonium perchlorate recorded at Au (1.5 mm). The sweep rate was 100 mV s⁻². The C₆₀/Ir and C₇₀/Ir films were grown under cyclic voltammetry conditions in acetonitrile/toluene (1:4, *v*.*v*) containing 0.10 M tetra(*n*-butyl)ammonium perchlorate, 0.25 mM C₆₀ or C₇₀, and 0.85 mM IrCl(CO)₂(*p*-toluidine)

Conclusions

So far, studies of formation and properties of co-polymers of fullerenes and transition metal complexes were focused on C₆₀-containing materials. In this paper, processes of formation and properties of C_{60}/M and C_{70}/M (M = Pt, Pd, Ir, Rh) are compared. For the same condition of electropolymerization, efficiency of C_{70}/M film formation is higher than the yield of C_{60}/M deposition. Also, processes related to the reduction of fullerene cage are more reversible for C₇₀ containing-films in comparison to the C_{60}/M layers. Voltammograms of C_{70}/M films in acetonitrile containing supporting electrolyte exhibit reduction peaks that are better defined and separated. Both differences in yield of electropolymerization and redox properties of C_{60}/M and C_{70}/M are probably related to structural differences of these films. Films containing C70 exhibit higher porosity in comparison to C₆₀ analogs. Therefore, transport of counterions during the electrode processes and solvent swelling are easier for C_{70}/M .

In fullerene/Pd and fullerene/Pt films, C_{60} or C_{70} cages are bounded through palladium or platinum atoms to form a polymeric network. These films are formed in potential range of metal complex reduction [31, 34]. Despite the similarity in the mechanism of film formation and film structure, both polymers exhibit different electrochemical properties. In the case of C_{60} /Pt and C_{70} /Pt systems, the polymeric phase deposited on the electrode surface inhibits the reduction processes at negative potentials. Also, charge transfer processes related to the reduction of fullerene cages in fullerene/Pt films are much slower and less reversible in comparison to the fullerene/Pd films.

In the case of fullerene/Rh and fullerene/Ir films, the fullerene cages are bridged by $Rh(CF_3CO_2)_2$ or $Ir(CO)_2$ moieties [30]. The voltammetric responses of these polymers in acetonitrile solution indicate a lower rate of charge transfer processes in comparison to fullerene/Pd films.

The spacer bridging fullerene cages also influences the yield of polymer formation. The highest yield of polymer formation was observed for fullerene/Rh systems. Yields of fullerene/Pt and fullerene/Ir electropolymerization were much lower in comparison to the two other studied systems.

A comparison of voltammograms of investigated films recorded in acetonitrile solution containing supporting electrolyte only indicates that fullerene/Pd and fullerene/ Rh can be considered as good electrochemical capacitors. In the case of rhodium films, a typical capacitance behavior, pseudorectangular cathodic profile of voltammograms, is observed only in low potential range, limiting its application for capacitors. Therefore, in this paper, capacitance studies were focused on C_{60} /Pd and C_{70} /Pd films. Studied polymers exhibit high pseudocapacitance. Specific capacitance of polymers increase with decrease in size of supporting electrolyte cations. C_{70} /Pd films exhibit better capacitance performance than films of C_{60} /Pd.

Values of specific capacitance reported in this paper are generally lower than what were given previously [38] for films deposited on smooth gold electrode. This effect is related to the differences in conditions of film formation. Results presented in this paper indicate that capacitance performance of the polymer depends on the amount of polymeric material deposited on the electrode surface. However, only a fraction of the film having direct contact with the solution takes place in electrode processes. Therefore, the fraction of electrochemically inactive material becomes bigger for thicker films, and specific capacitance decreases.

Acknowledgment Support from the Polish State Committee for Scientific Research (grant N204 3747 33) is gratefully acknowledged.

References

- Novak P, Muller K, Santhanam KSV, Haas O (1997) Chem Rev 97:207
- 2. Burke A (2000) J Power Sources 91:37
- 3. Wrighton MS (1986) Science 231:32
- 4. Fortier G, Brassard E, Belanger D (1990) Biosens Bioelectron 5:473

- Yoshima H, Kobayashi M, Lee KB, Chung D, Heeger AJ, Wudl F (1987) J Electrochem Soc 134:46
- 6. Wudl F (2002) J Mater Chem 12:1959
- 7. Chen Y, Huang ZE, Cai RF, Yu BC (1998) Eur Polym J 34:137
- 8. Echegoyen L, Echegoyen LE (1998) Acc Chem Res 31:593
- 9. Winkler K, Balch AL, Kutner W (2006) J Solid State Electrochem 10:761
- 10. Gunes S, Neugebauer H, Sariciftci NS (2007) Chem Rev 107:1324
- 11. Rao CNR, Seshadri R, Govindaraj A, Sen R (1995) Mater Sci Eng R15:209
- Wilson LJ, Cagle DW, Thrash TP, Kennel SJ, Mirzadeh S, Alford JM, Ehrhardt GJ (1999) Coord Chem Rev 190–192:199
- 13. Yeretzian C, Hansen K, Diedrich FN, Whetten RL (1992) Nature 359:44
- 14. Ito A, Morikawa T, Takahashi T (1993) Chem Phys Lett 211:333
- 15. Zhou P, Dong ZH, Rao A, Ecklund PC (1993) Chem Phys Lett 211:337
- Takahashi N, Dock H, Matsuzawa N, Ata M (1993) J Appl Phys 74:5790
- Yamawaki H, Yoshida M, Kakadate Y, Usuba S, Yokoi H, Fujiwara S, Aoki K, Ruoff R, Malhorta R, Lorents DC (1993) J Phys Chem 97:11161
- Rao AM, Zhou P, Wang KA, Hager GT, Holden JM, Wang Y, Lee WT, Bi XX, Eklund PC, Cornett DC, Duncan MA, Amster IJ (1993) Science 250:955
- 19. Loy DA, Assink RA (1992) J Am Chem Soc 114:3977
- Bunker CE, Lawson GE, Sun YP (1995) Macromolecules 28:7959
- 21. Cao T, Webber SE (1996) Macromolecules 29:3826
- 22. Ford WT, Graham TD, Mourey HT (1997) Macromolecules 30:6422
- 23. Hirsch A, Li Q, Wudl F (1991) Angew Chem Int Ed 30:1309
- 24. Sun YP, Liu B, Lawson GE (1997) Photochem Photobiol 66:301
- 25. Hawker CJ (1994) Macromolecules 27:4836
- 26. Rubin Y, Khan S, Freedberg D, Yeretzian C (1993) J Am Chem Soc 118:344
- 27. Fedurco M, Costa DA, Balch AL, Fawcett WR (1995) Angew Chem Int Ed Engl 34:194
- Winkler K, Costa DA, Balch AL, Fawcett WR (1995) J Phys Chem 99:17431
- Krinichnaya EP, Moravsky AP, Efimov O, Sobczak JW, Winkler K, Kutner W, Balch AL (2005) J Mater Chem 15:1468
- Balch AL, Costa DA, Winkler K (1998) J Am Chem Soc 120:9614
- Winkler K, de Bettencourt-Dias A, Balch AL (1999) Chem Mater 11:2265
- 32. Winkler K, de Bettencourt-Dias A, Balch AL (2000) Chem Mater 12:1386
- Winkler K, Noworyta K, Kutner W, Balch AL (2000) J. Electrochem Soc 147:2597
- Hayashi A, de Bettencourt-Dias A, Winkler K, Balch AL (2002) J Mater Chem 12:2116
- Plonska ME, de Bettencourt-Dias A, Balch AL, Winkler K (2003) Mater Chem 15:4122
- Plonska ME, Makar A, Winkler K, Balch AL (2004) Pol J Chem 78:1431
- Winkler K, Plonska-Brzezinska ME, Gadde S, D'Souza F, Balch AL (2006) Electroanalysis 18:841
- Winkler K, Grodzka E, D'Souza F, Balch AL (2007) J Electrochem Soc 154:K1
- 39. Forster D (1969) Inorg Nucl Chem Lett 5:433
- Winkler K, Noworyta K, Sobczak JW, Wu CT, Chen LC, Kutner W, Balch AL (2003) J Mater Chem 23:518
- Ward MD (1995) In: Rubinstein I (ed) Physical Electrochemistry. Marcel Dekker, New York Basel Hong Kong
- 42. Grodzka E, Nieciecka M, Winkler K (2008) J Solid State Electrochem 12:215